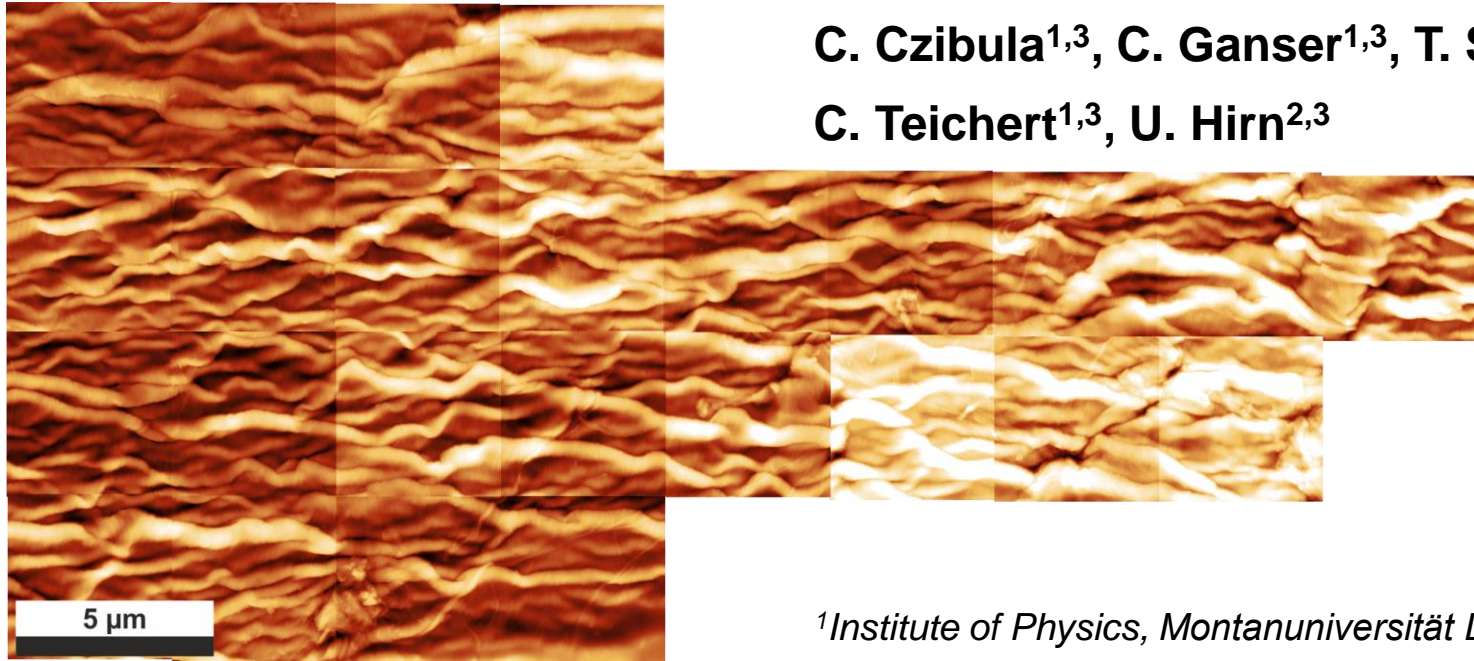


# TRANSVERSE VISCOELASTIC PROPERTIES OF PULP FIBERS INVESTIGATED WITH ATOMIC FORCE MICROSCOPY

C. Czibula<sup>1,3</sup>, C. Ganser<sup>1,3</sup>, T. Seidlhofer<sup>2,3</sup>,  
C. Teichert<sup>1,3</sup>, U. Hirn<sup>2,3</sup>



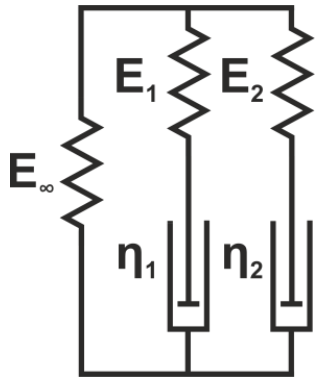
**ПРОБЛЕМЫ МЕХАНИКИ  
ЦЕЛЛЮЛОЗНО-БУМАЖНЫХ  
МАТЕРИАЛОВ**  
Архангельск, 11.09.2019

*<sup>1</sup>Institute of Physics, Montanuniversität Leoben, Austria*

*<sup>2</sup>Institute for Paper, Pulp and Fibre Technology, Graz University of Technology, Austria*

*<sup>3</sup>Christian Doppler Laboratory for Fiber Swelling and Paper Performance, Graz University of Technology, Austria*

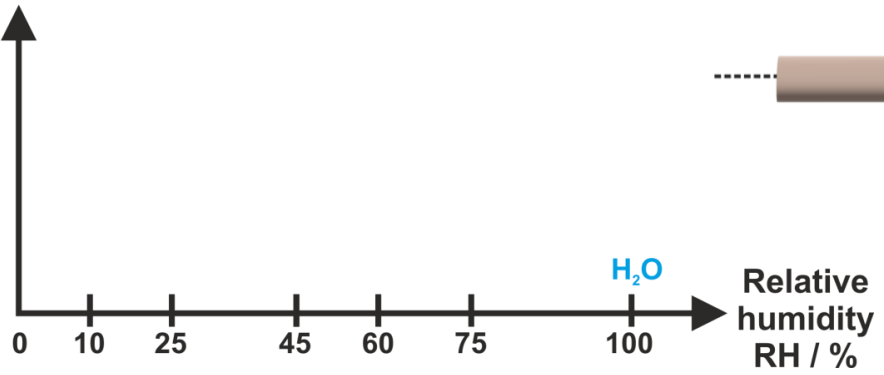
# Motivation



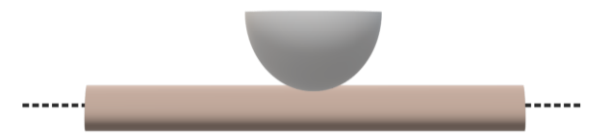
Viscoelasticity

Viscoelastic properties

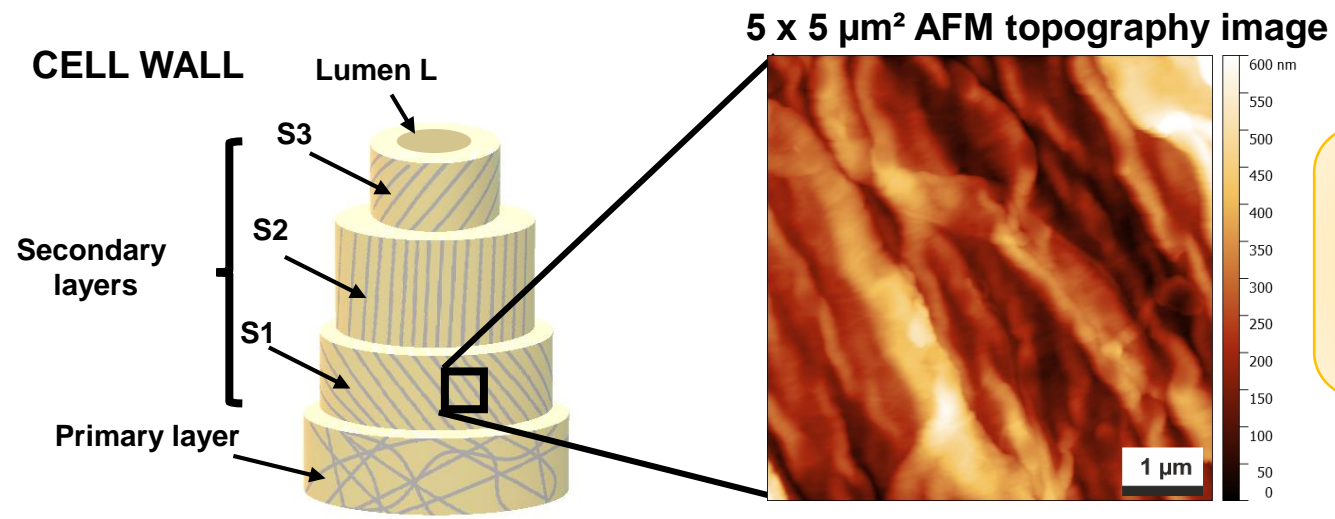
$E_i, \eta_i, \tau_i$



Transverse fiber direction



## Hierarchical structure of pulp fibers

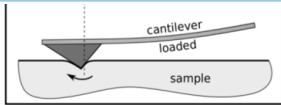


Challenging:  
rough surface  
& anisotropic  
properties

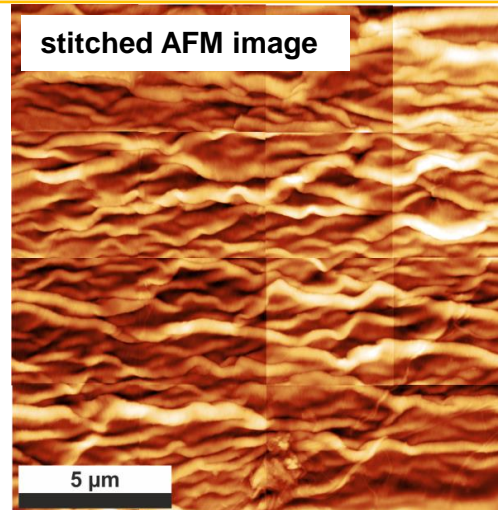
# Background

Influence of relative humidity

AFM Nanoindentation



Transversal viscoelastic properties of single cellulose fibers with AFM methods



Reduced modulus  $E_r$  & Hardness  $H$

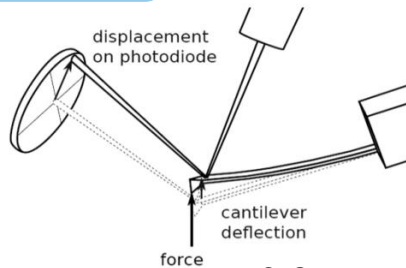
Validation of the method with Polycarbonate (PC) & Polymethylmethacrylate (PMMA) and comparison with tensile testing and nanoindentation

C. Ganser et al., *Soft Matter* 14 (2018) 140-150.

Past experience

Adhesion force  $F_{ad}$

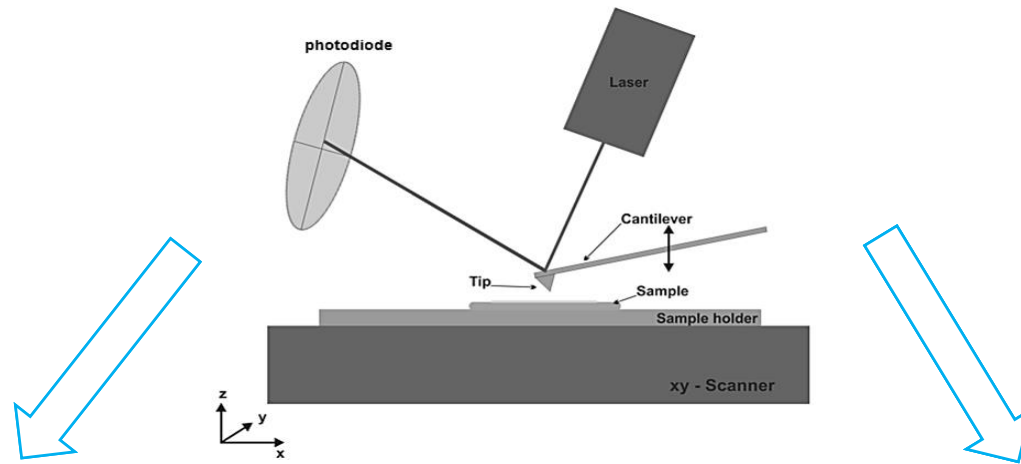
AFM force measurements



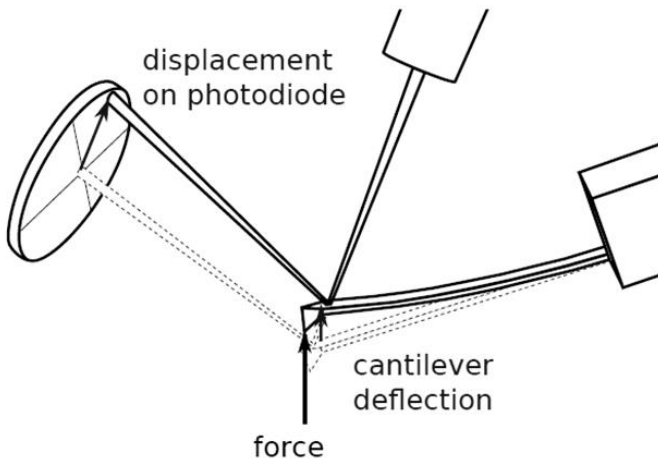
Chemical contrast with functionalized probes

C. Ganser et al., *Holzforschung* 70 (2016) 1115-1123.

# Atomic force microscopy (AFM)

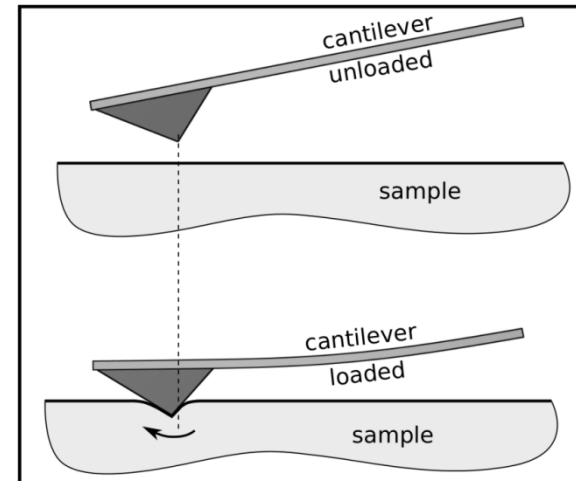


## Force measurements



- Cantilever deflection  $\leftrightarrow$  Force
- Adhesion, mechanical properties

## AFM nanoindentation (AFM-NI)

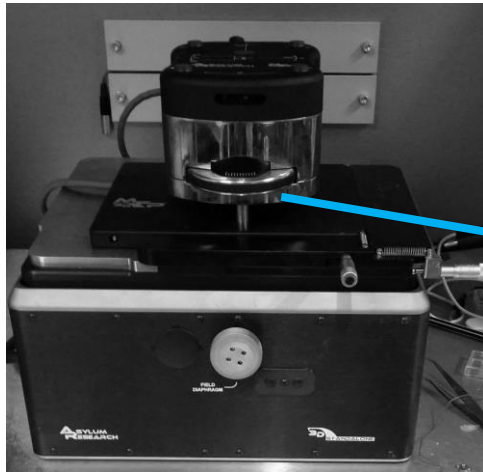
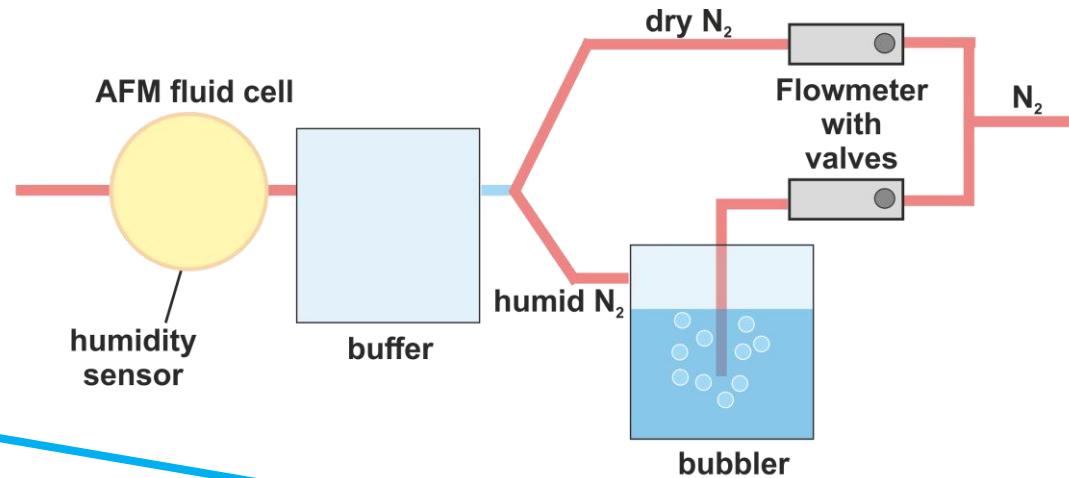


- Lateral movement of the tip
- Force application not purely vertical

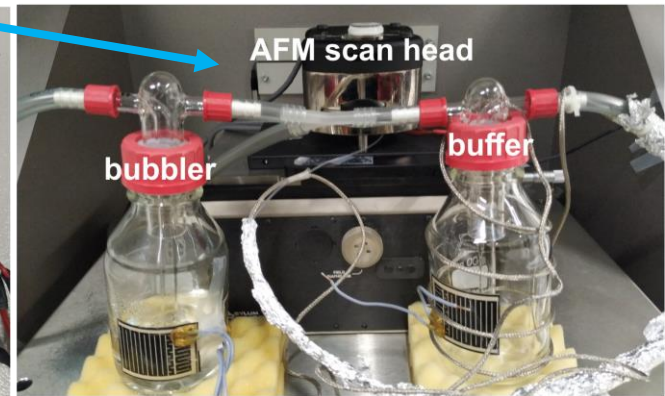
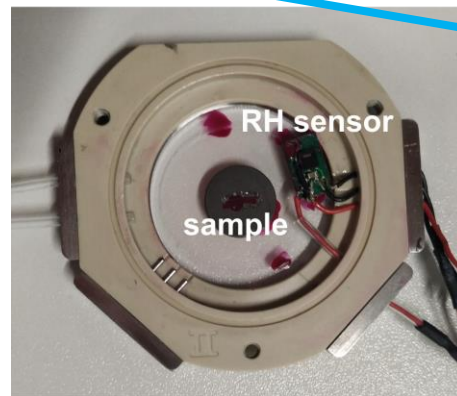


# Experimental setup

## Relative humidity (RH) setup



**AFM: Asylum  
Research MFP 3D**



**With the setup it is possible to measure between  
~10 and ~80 % RH for the viscoelastic experiments.**

C. Ganser et al., *Holzforschung* **70** (2016) 1115-1123.

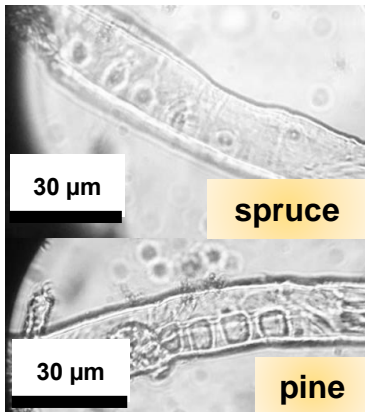
C. Ganser, C. Teichert, in "Applied Nanoindentation of Advanced Materials", Ed. A. Tiwari, pp 247-267, Wiley, 2017.

# Experimental

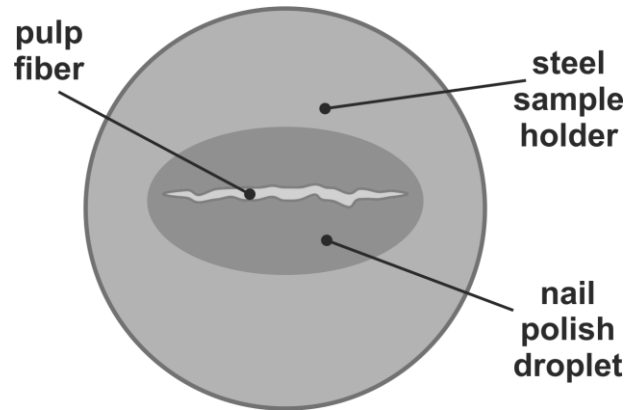
## Samples

### Pulp fibers

Unbleached, unrefined kraft pulp fibers; mixture of spruce and pine (4 spruce & 2 pine fibers)  
 $d \sim 20 - 30 \mu\text{m}$ ,  $l \sim 3 - 5 \text{ mm}$



## Sample preparation

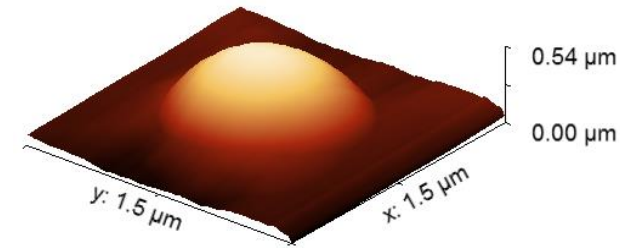


W. J. Fischer et al. *Cellulose* 21 (2014) 237-249.

## AFM probe

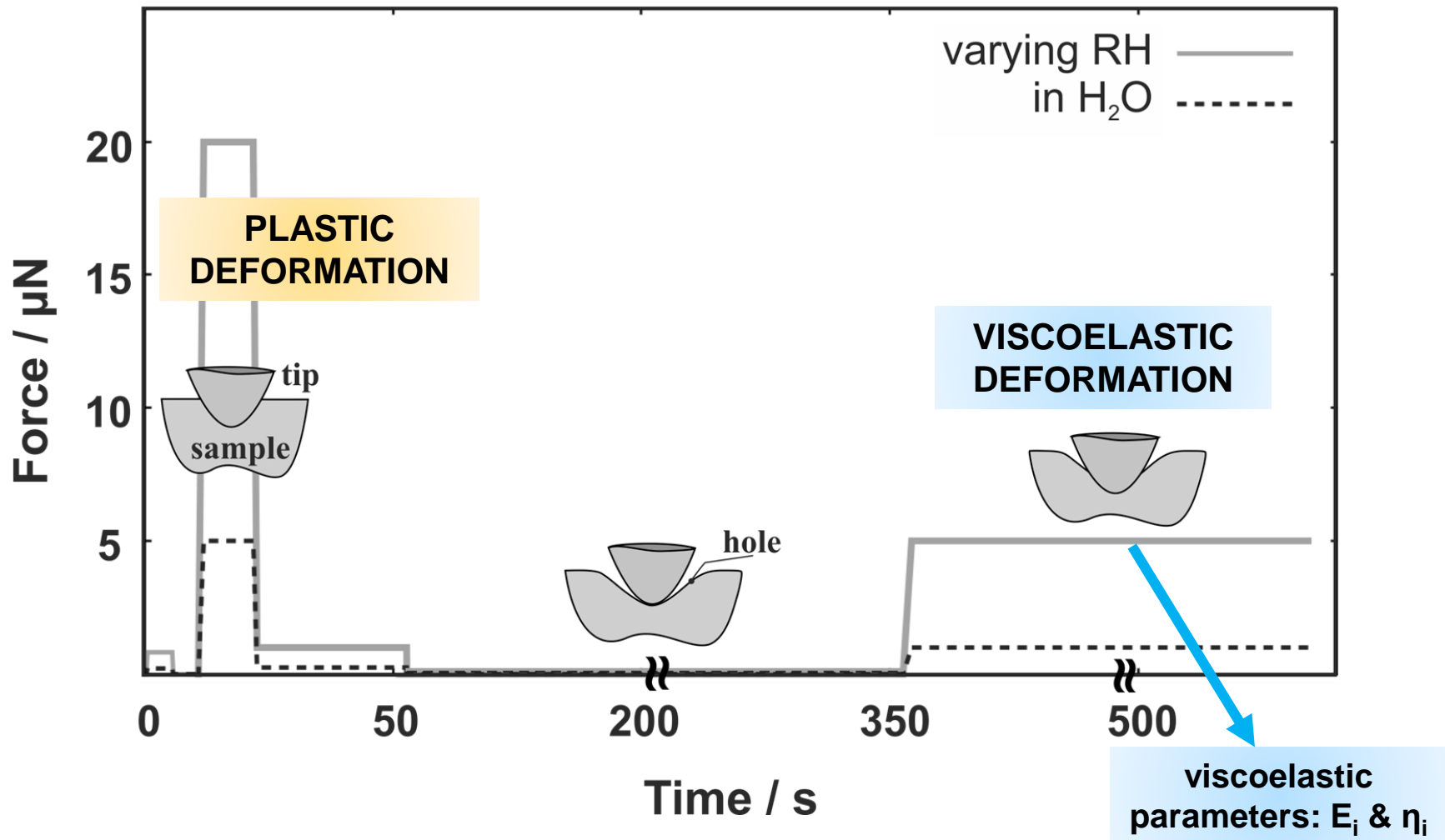
### Large radius hemispherical probe

tip radius  $\sim 300 \text{ nm}$   
spring constant of the cantilever  $\sim 250 \text{ N/m}$   
(Team Nanotec)



- Tip geometry is well known
  - Larger radius  $\rightarrow$  higher forces possible
- Drawback:** reduced lateral resolution

# Load schedule



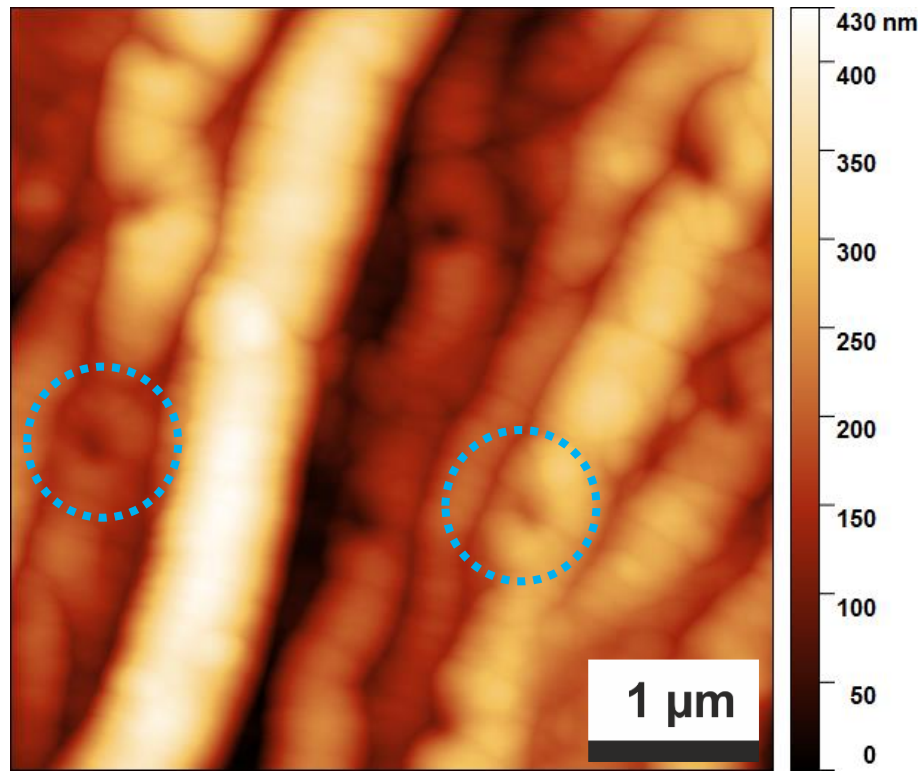
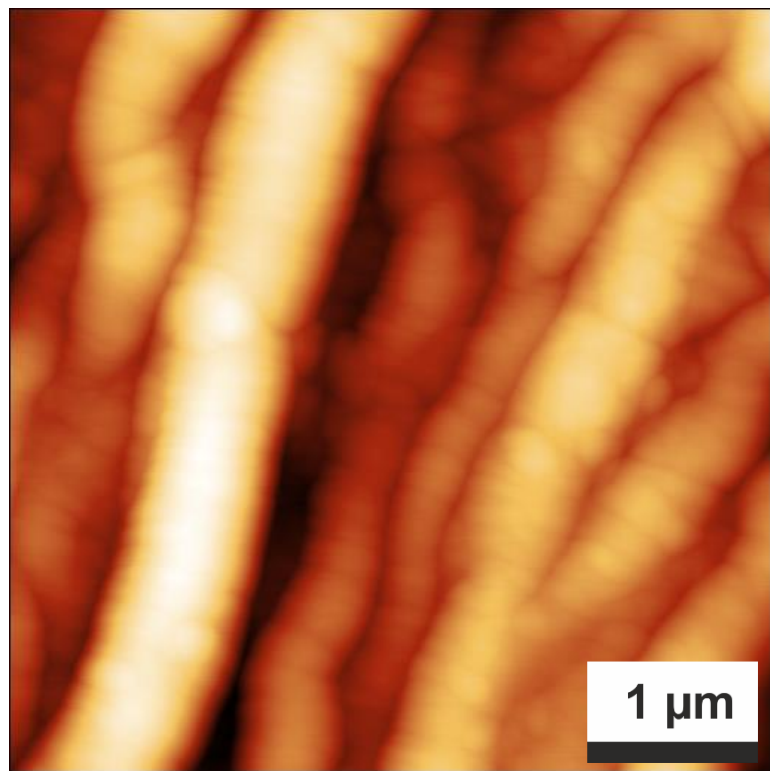
# Measurements – 5 x 5 $\mu\text{m}^2$ topography images



Cantilever  
orientation  
along the long  
fiber axis

Before AFM-NI

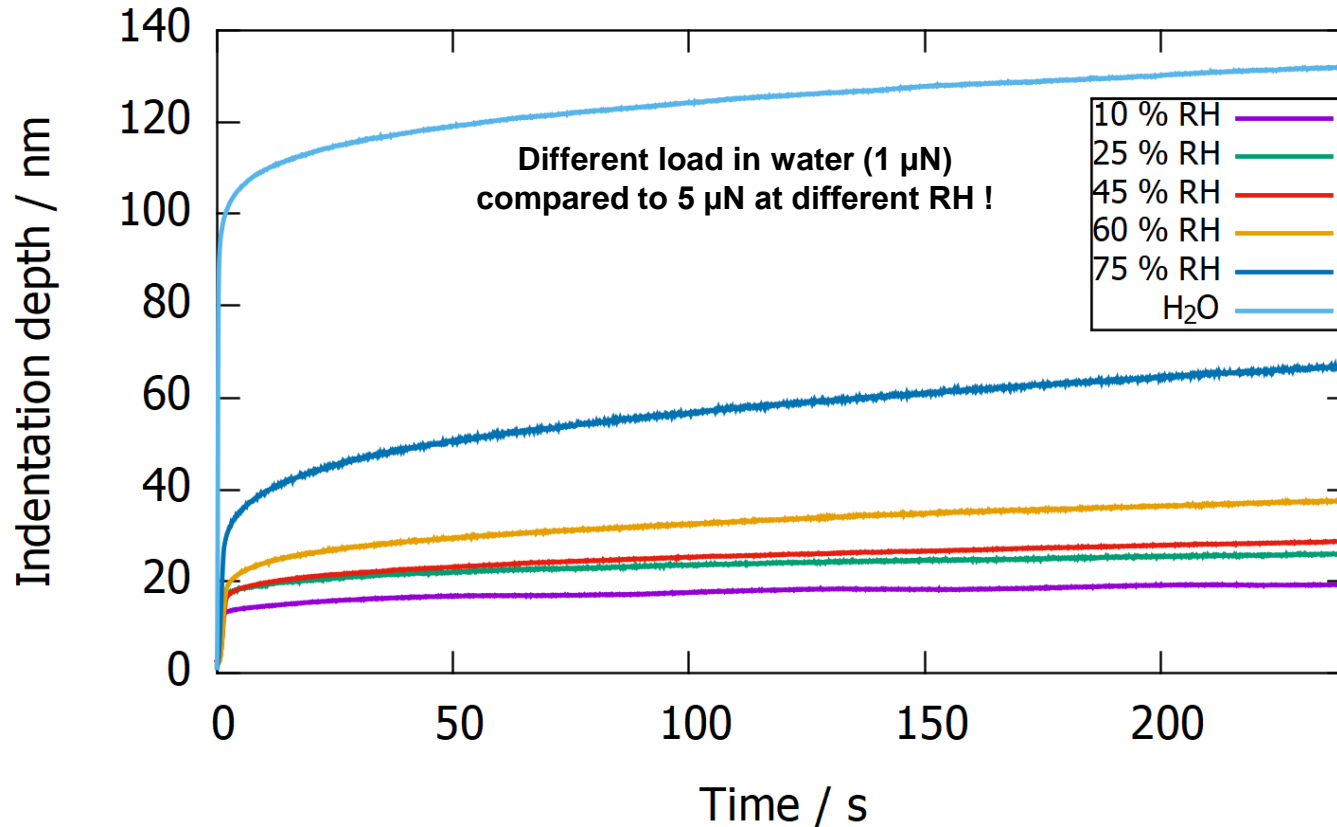
After AFM-NI





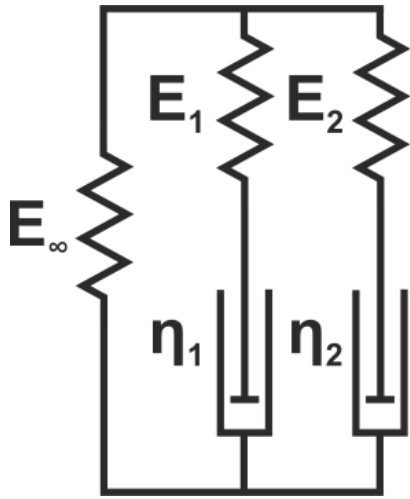
# Experimental creep curves

*Averaged curves from 6 different unbleached pulp fibers*



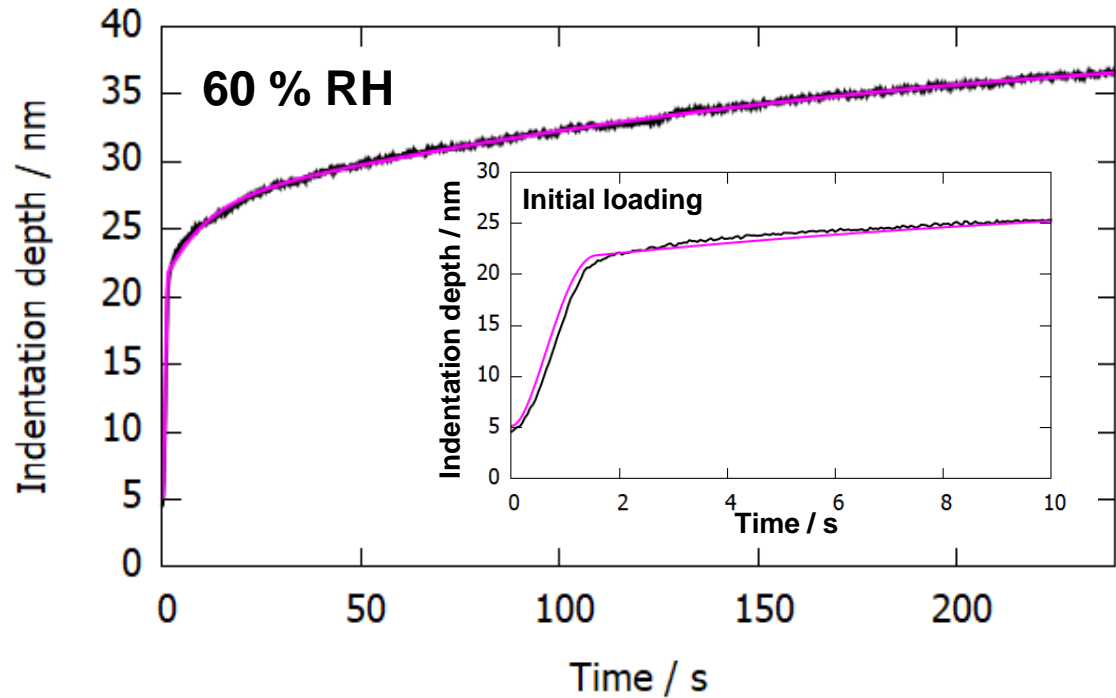
**Indentation depth** and **slope** increase with increasing **RH** as expected.

# Generalized Maxwell model of order 2 (GM2)



$$\sigma = A\varepsilon + B\dot{\varepsilon} + C\ddot{\varepsilon} - D\dot{\sigma} - E\ddot{\sigma}$$

Experimental (black line) & fit (pink line) curve for a pulp fiber

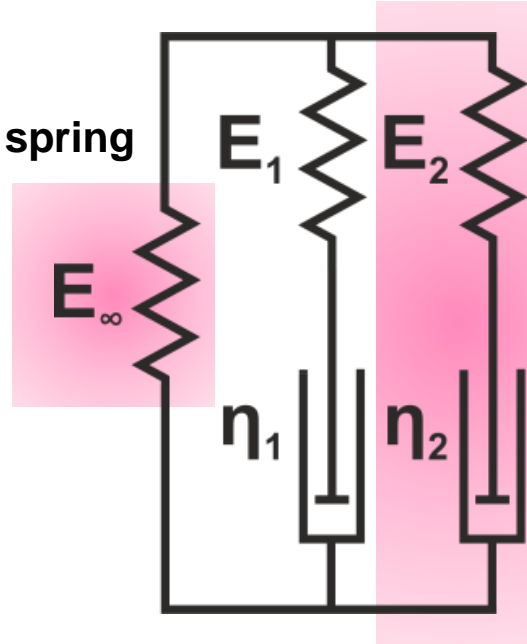


H.T. Banks, S. Hu, Z.R. Kenz. *A Brief Review of Elasticity and Viscoelasticity*. N. C. S. University.  
 Rayleigh, 2010.  
 D. Roylance. *Engineering Viscoelasticity*. Massachusetts Institute of Technology. Cambridge, 2001.

# Dissecting the GM2 model

2 Maxwell elements

single spring



*infinitely slow loading:*

$$E_{\infty}$$

*infinitely fast loading:*

$$E_0 = E_{\infty} + E_1 + E_2$$

*Relaxation time*

$$\tau_i = \frac{\eta_i}{E_i}$$

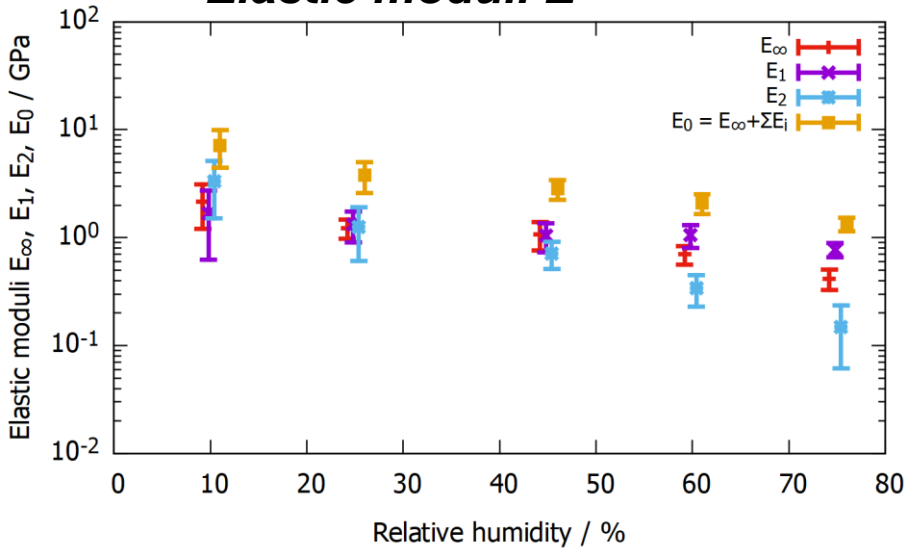
*ratio between viscosity and stiffness*

## Viscoelastic characterization:

- *Elastic parameters:*  $E_{\infty}, E_1, E_2, E_0$
- *Viscous parameters:*  $\eta_1, \eta_2$
- *Relaxation behavior:*  $\tau_1, \tau_2$

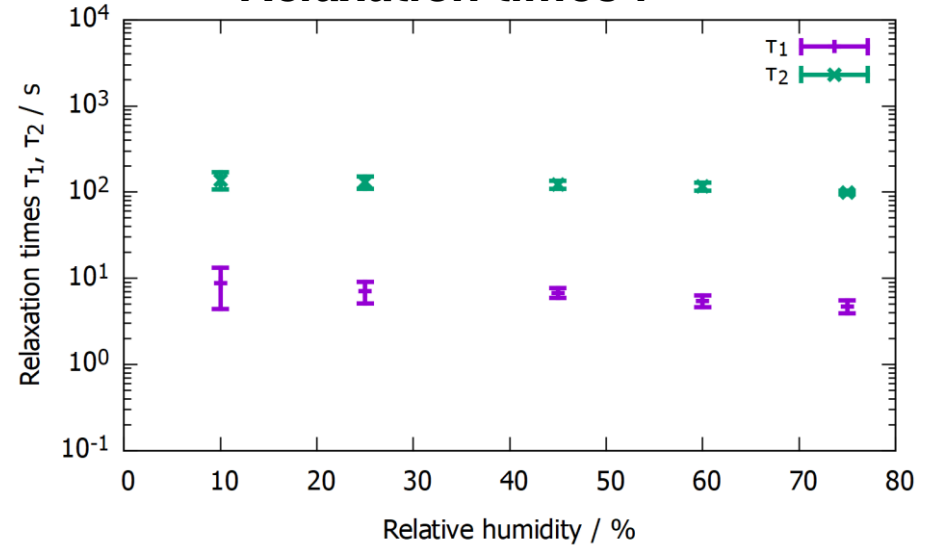
# Results for unbleached pulp fibers at different RH

## Elastic moduli $E$

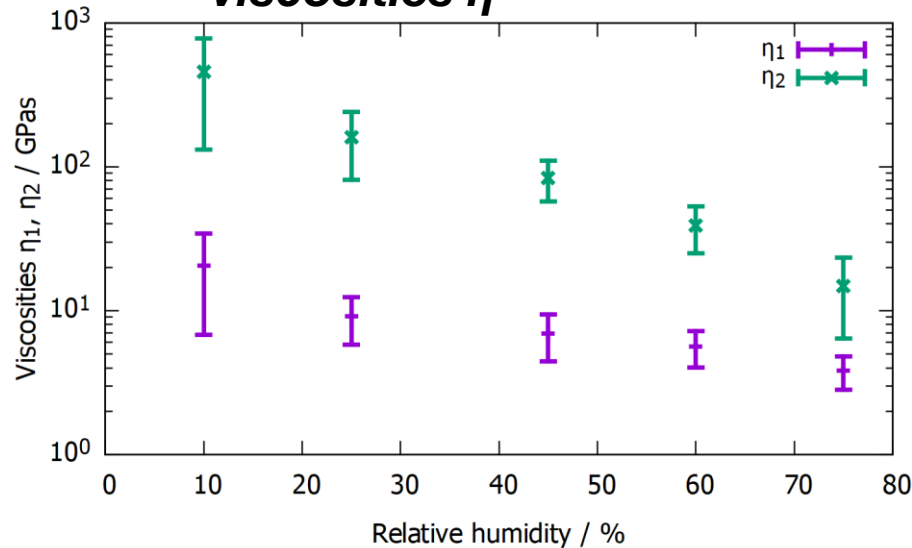


6 unbleached fibers (4 spruce and 2 pine fibers)

## Relaxation times $\tau$



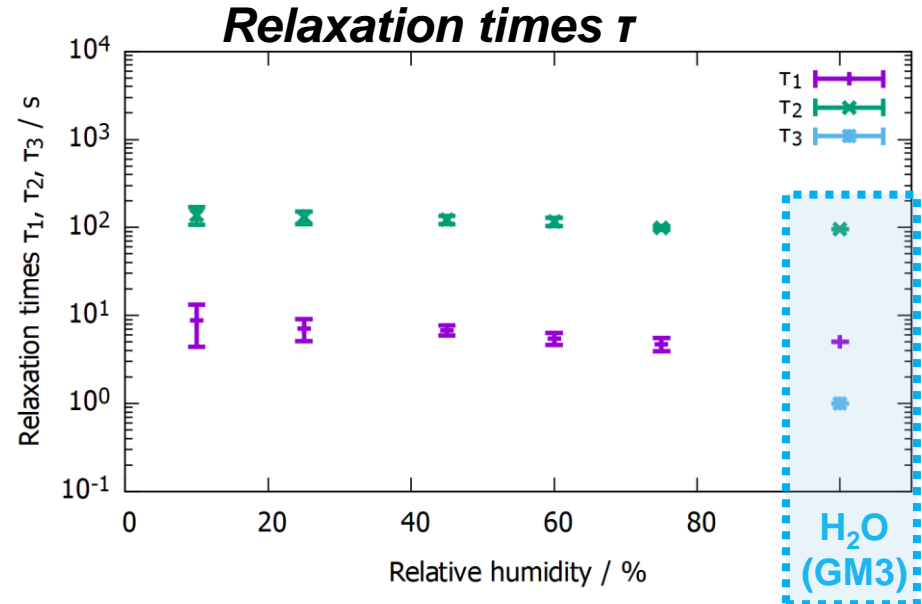
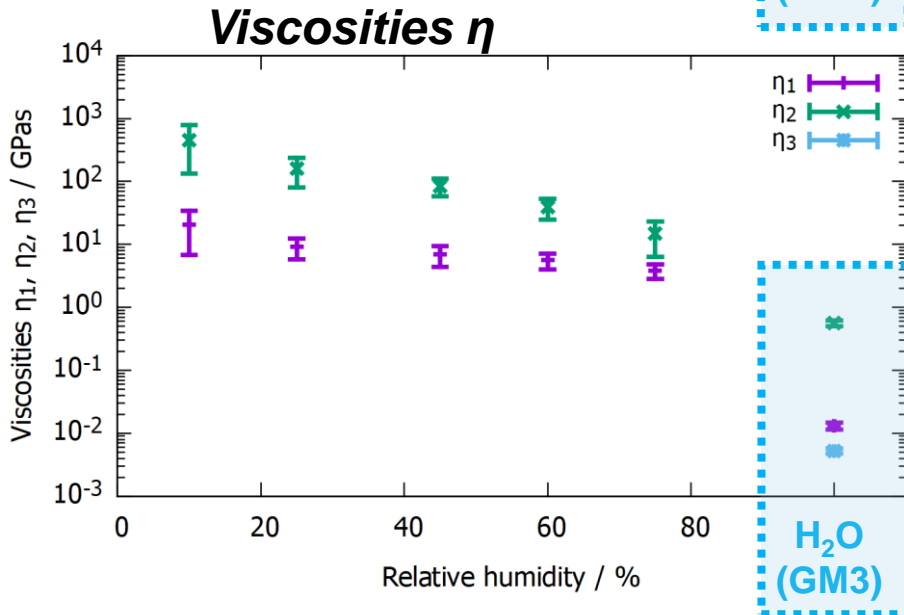
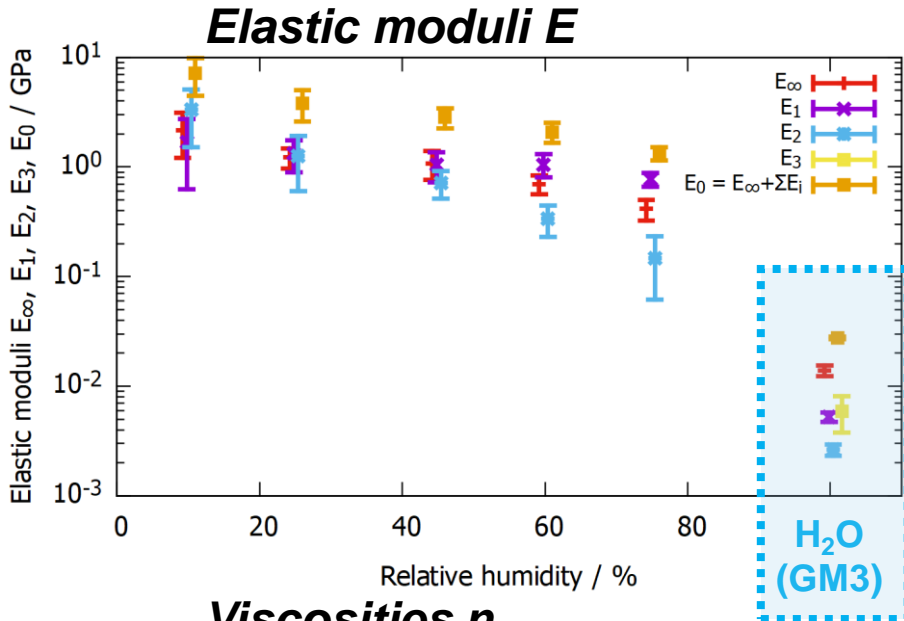
## Viscosities $\eta$



Both, elastic moduli and viscosities show a gradual decrease with increasing RH. The relaxation times show little change.

# Results for unbleached pulp fibers

6 unbleached fibers (4 spruce and 2 pine fibers)



Both, elastic moduli and viscosities show a decrease with increasing RH. In water the decrease is highest.

The elastic moduli and viscosities decrease by a factor of 2 and 4, respectively.

The relaxation times show little change, only at higher RH and in water a third one was added.

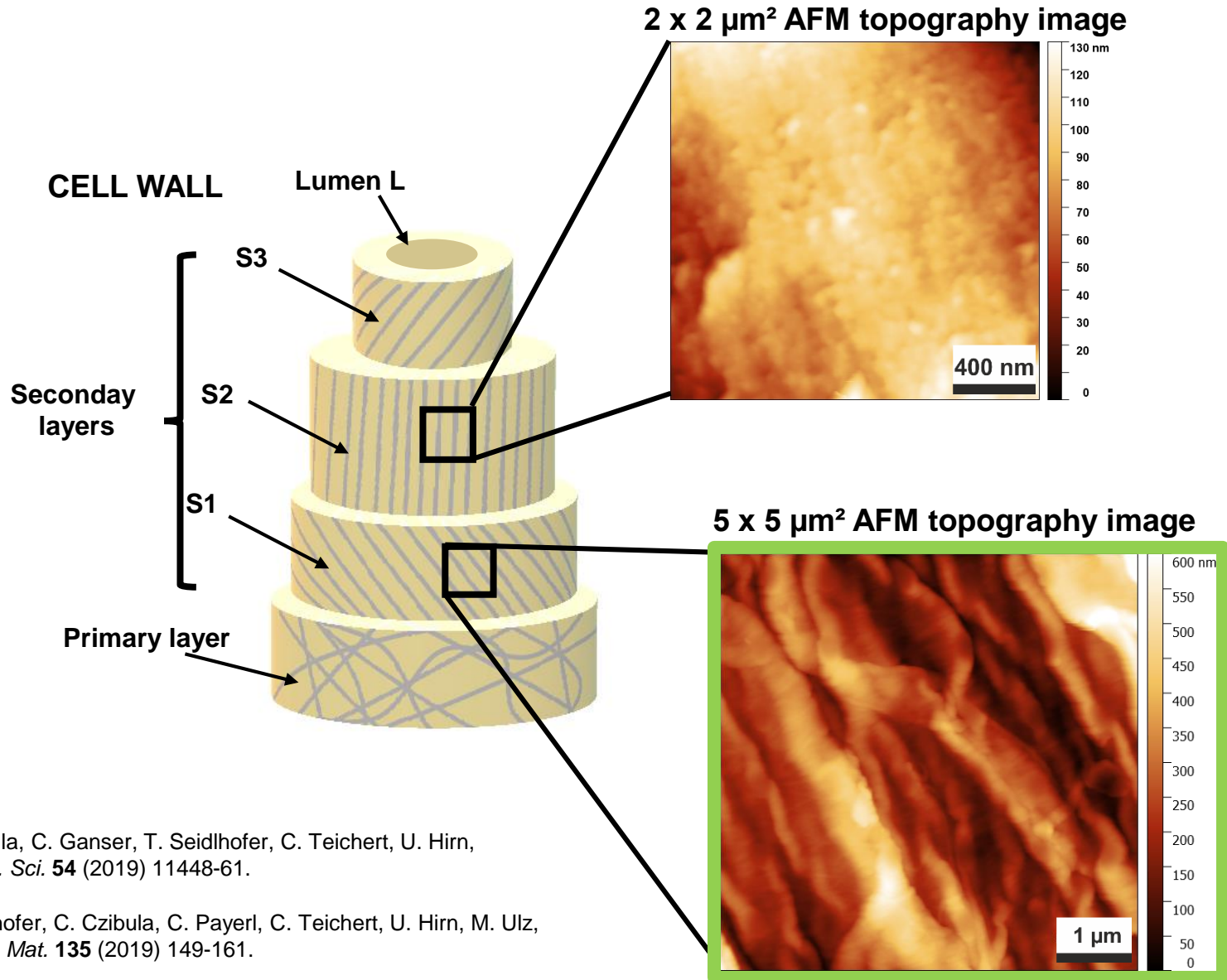


# Conclusions

- An AFM based method to investigate the **viscoelastic properties of pulp fibers** was presented.
- Both, **elastic moduli** and **viscosities** show a **decrease with increasing RH**. The **relaxation behavior** stays **constant**.
- **In water**, a **GM3 model** is needed to fit the data properly (hydrogel-like behavior, poroelastic approach).

# Outlook:

microtome cuts to access the S2 layer



C. Czibula, C. Ganser, T. Seidlhofer, C. Teichert, U. Hirn,  
*J. Mater. Sci.* **54** (2019) 11448-61.

T. Seidlhofer, C. Czibula, C. Payerl, C. Teichert, U. Hirn, M. Ulz,  
*J. Mech. Mat.* **135** (2019) 149-161.

# THANKS FOR YOUR ATTENTION !!



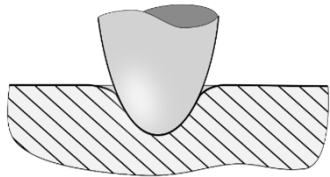
5 μm





# Theory behind data analysis

## Contact mechanics model



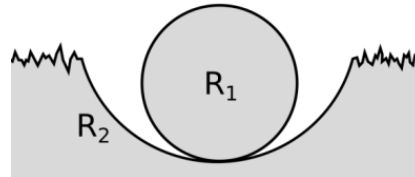
## Johnson-Kendall-Roberts (JKR)

### Assumptions:

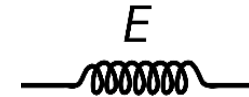
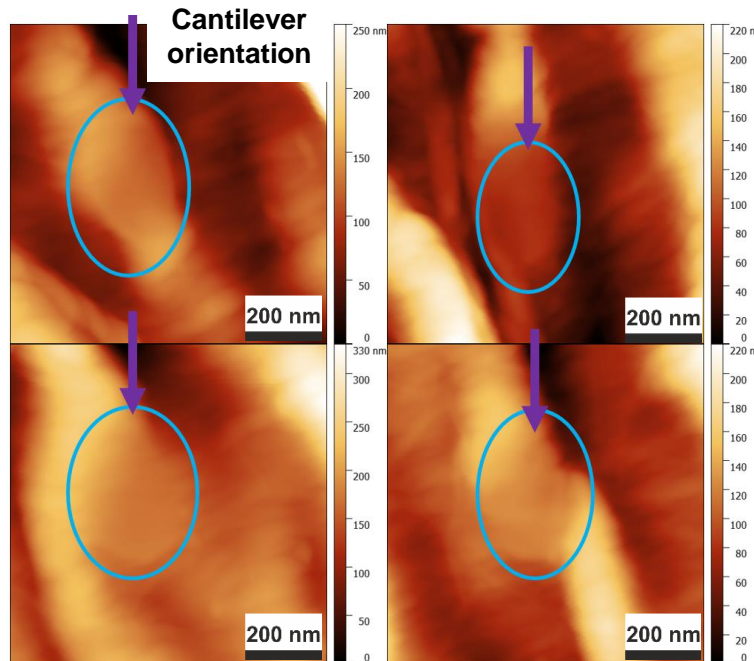
- Tip is a sphere
- Sample: plane, sphere, hole
- Topography: smooth
- Regime: elastic
- Adhesion: **YES**

K.L. Johnson, K. Kendall and A.D. Roberts,  
*Proc. R. Soc. Lond. A*, **324** (1971), 301-313.

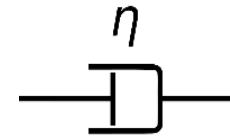
## How to define the tip-surface contact ?



## Contact in a hole

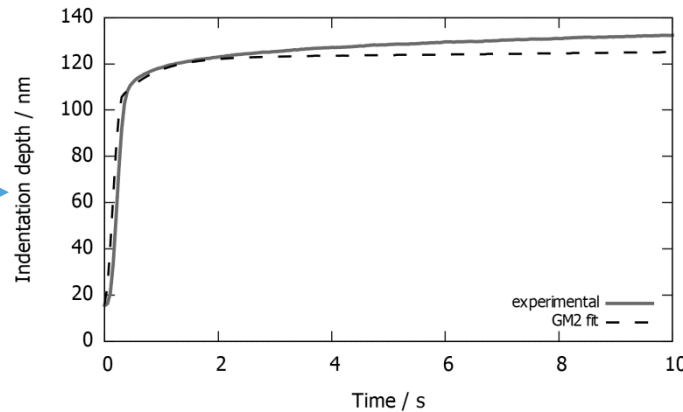
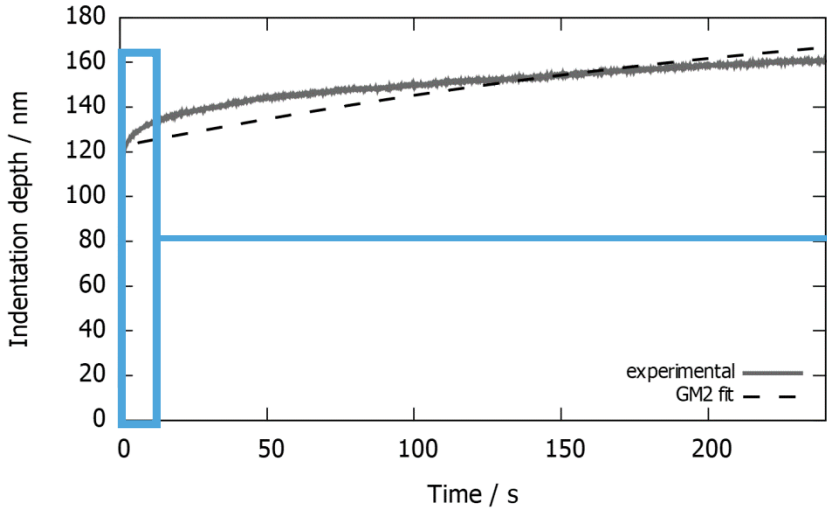
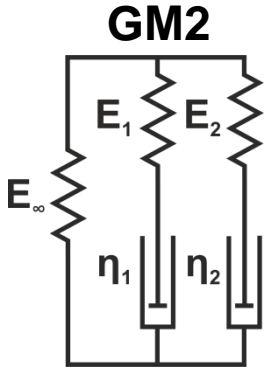


## + Viscoelastic model

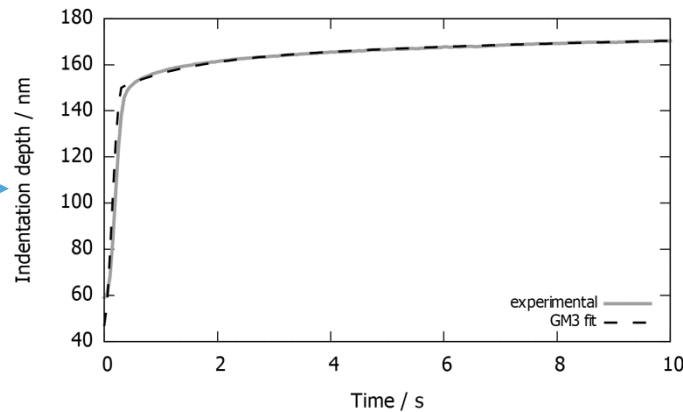
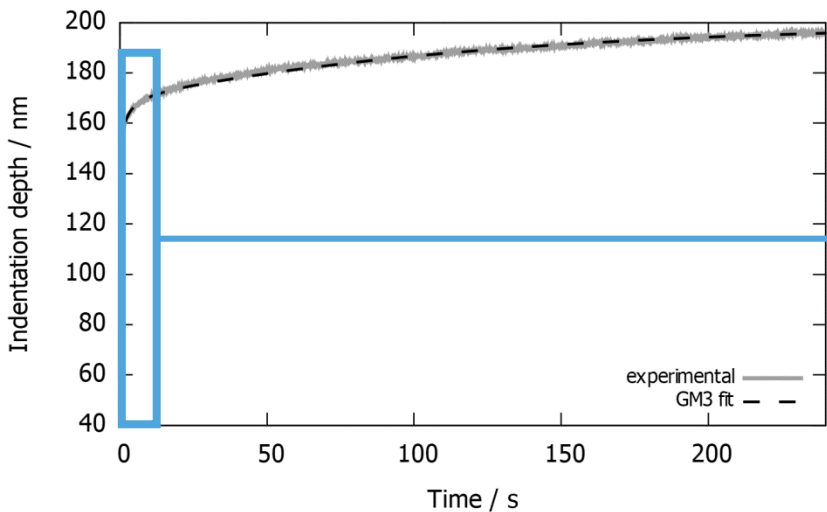
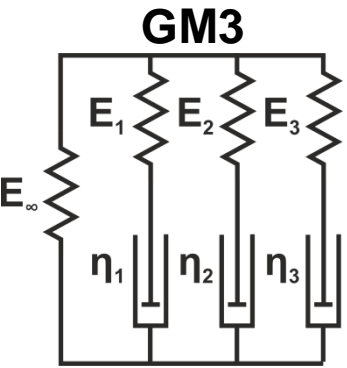




# Comparison of GM2 and GM3 fits for H<sub>2</sub>O measurements



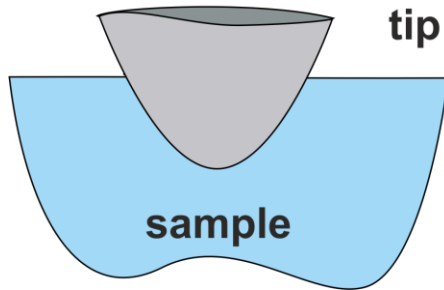
Introduction of a third Maxwell element with fixed relaxation times:  $\tau_1 = 1$  s,  $\tau_2 = 5$  s,  $\tau_3 = 100$  s



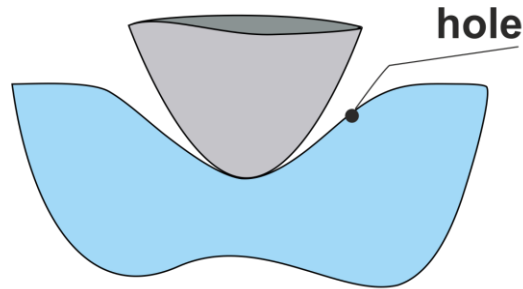
The GM3 model fits the experimental curves from the water measurements much better.

# Overcoming surface roughness

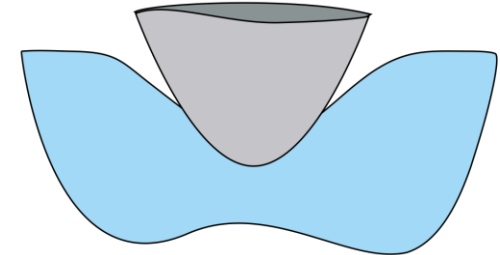
(I)  
Plastic  
deformation



(II)  
After  
plastic deformation

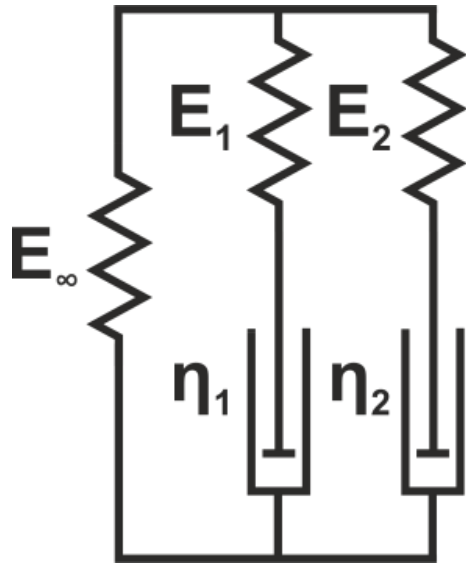


(III)  
Viscoelastic  
deformation

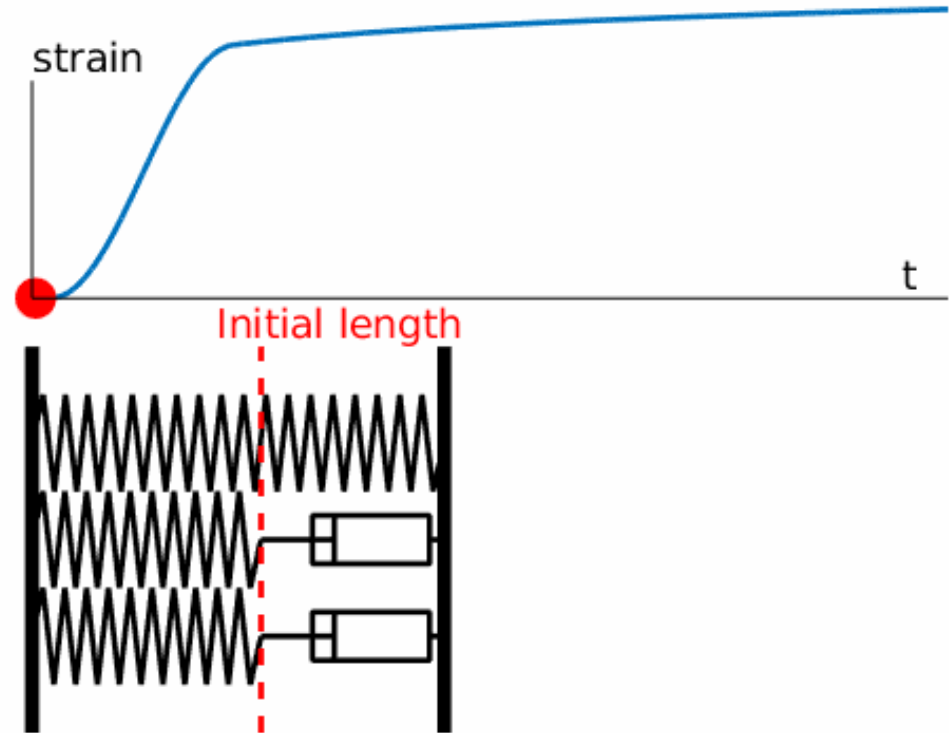


- surface roughness reduced
- defined contact area  
→ contact in a hole

# Generalized Maxwell order 2 (GM2) model

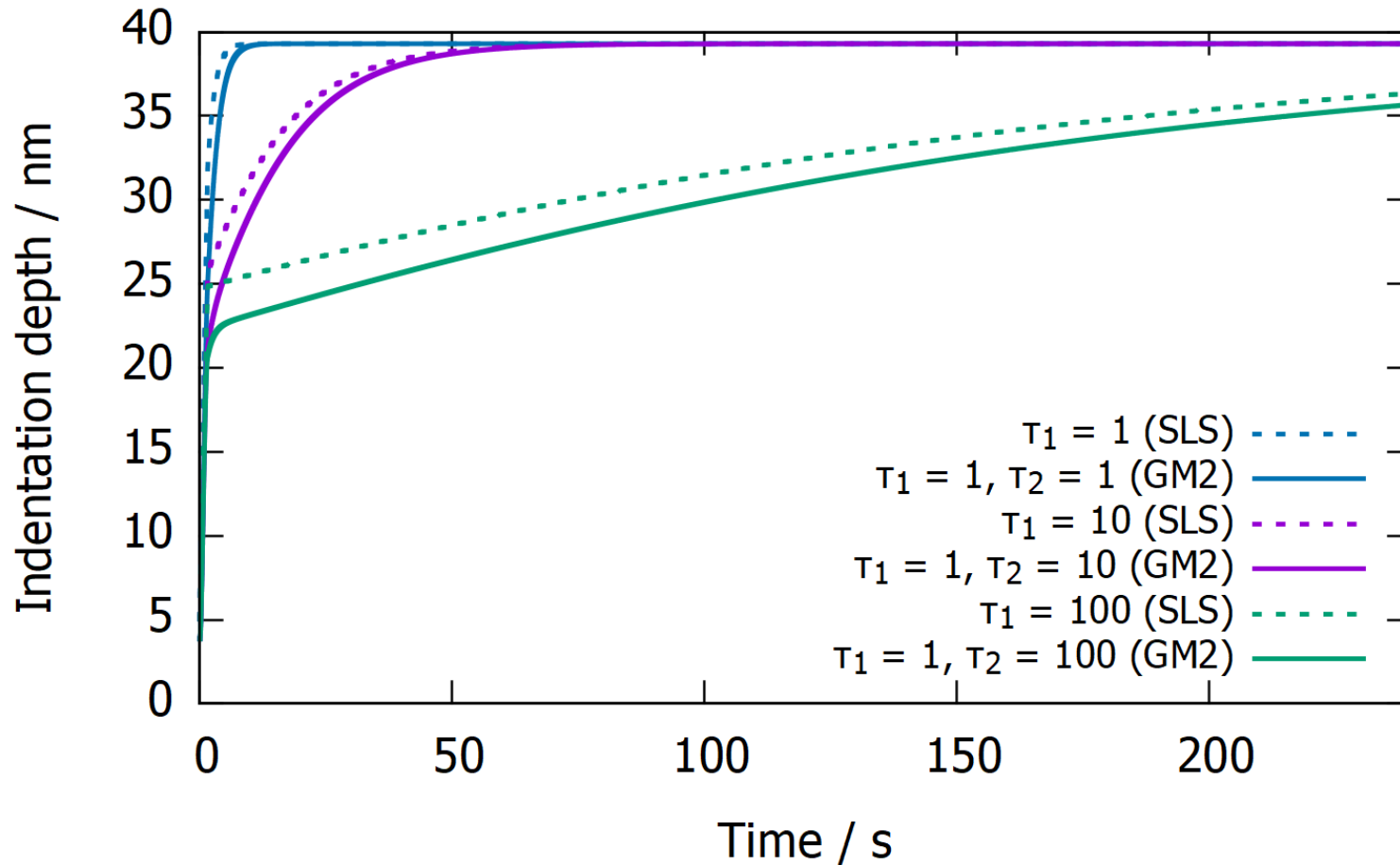


$$\sigma = A\varepsilon + B\dot{\varepsilon} + C\ddot{\varepsilon} - D\dot{\sigma} - E\ddot{\sigma}$$



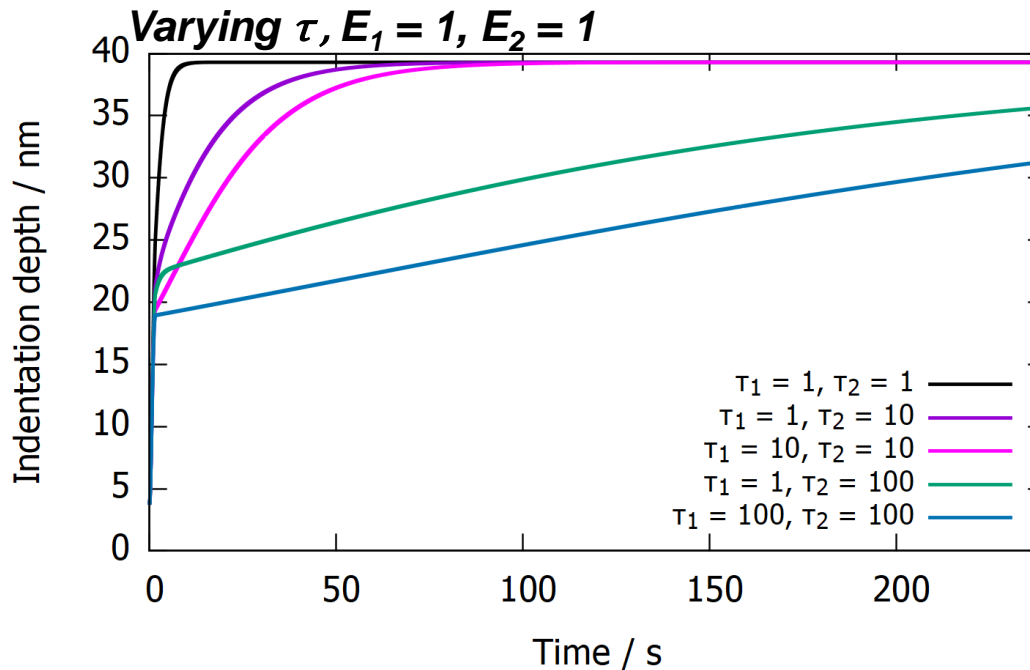
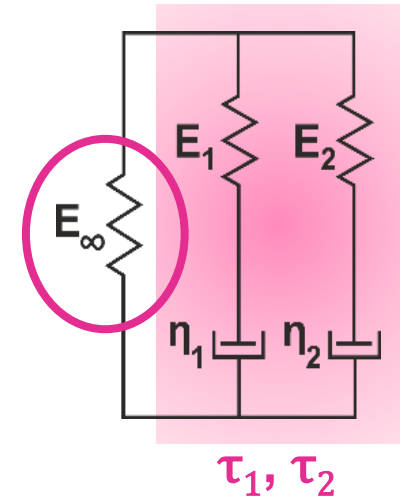
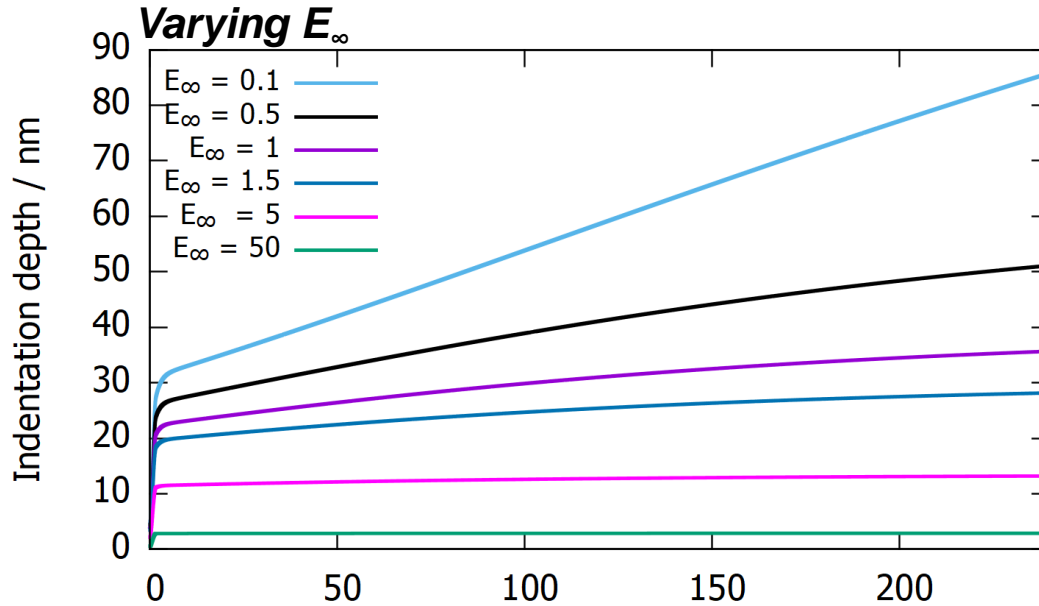
Courtesy by T. Seidhofer

# Comparison SLS & GM2



**Comparison between the SLS and GM2 model show that the additional Maxwell element influences the slope/form of the curves.**

# GM2 model



The form and depth of the indentation curve is mainly influenced by  $E_\infty$ .

The Maxwell elements are influencing mostly the initial slope and form of the indentation curve.



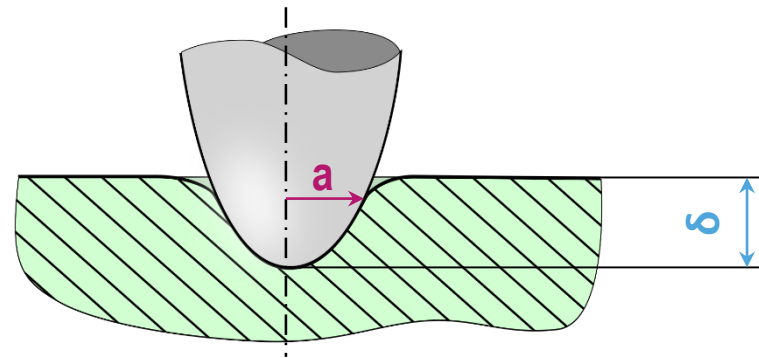
# Contact mechanics model

## Johnson-Kendall-Roberts (JKR)

A contact mechanics model is applied to convert force and indentation depth into stress and strain.

### JKR - Assumptions:

- **Tip:** sphere
- **Sample:** plane, sphere, hole
- **Topography:** smooth
- **Regime:** elastic
- **Adhesion: YES**

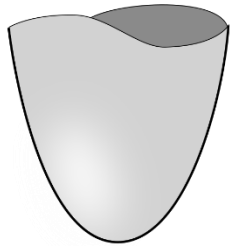


	contact radius $a$	indentation depth $\delta$	stress $\sigma$	strain $\varepsilon$
<b>JKR</b>	$a^3 = \frac{3R}{4E} \tilde{F}$	$\delta = \frac{a^2}{R} - \left( \frac{4a F_a}{3RE} \right)^{\frac{1}{2}}$	$\sigma = \frac{\Delta(\tilde{F})}{R\delta} \left[ \frac{3}{8} \tilde{F} - \frac{1}{2} (\tilde{F} F_a)^{\frac{1}{2}} \right]$	$\varepsilon = \left( \frac{\delta}{R} \right)^{\frac{1}{2}} \frac{1}{\Delta(\tilde{F})^{\frac{1}{2}}} \left[ \frac{1}{2} - \left( \frac{F_a}{\tilde{F}} \right)^{\frac{1}{2}} \right]$

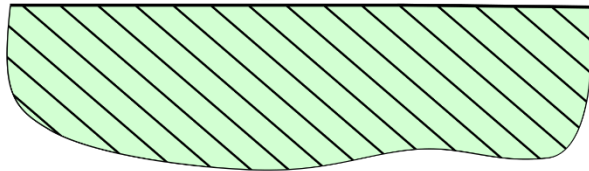
K.L. Johnson, K. Kendall and A.D. Roberts, *Proc. R. Soc. Lond. A*, **324** (1971), 301-313.

# Contact mechanics in a hole

(I)

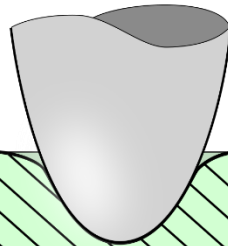


$F = 0$

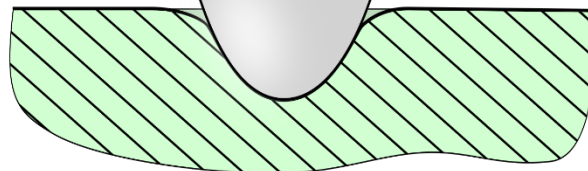


I. undeformed, virgin surface

(II)

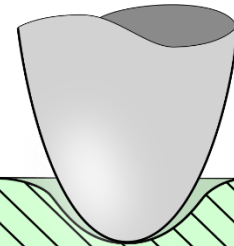


$F = F_{\max}$

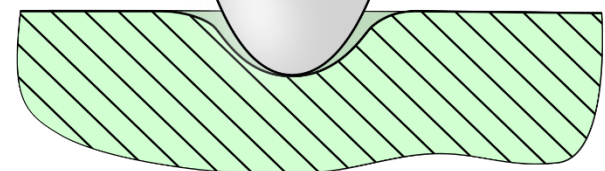


II. plastic deformation under maximum load

(III)



$F \approx 0$



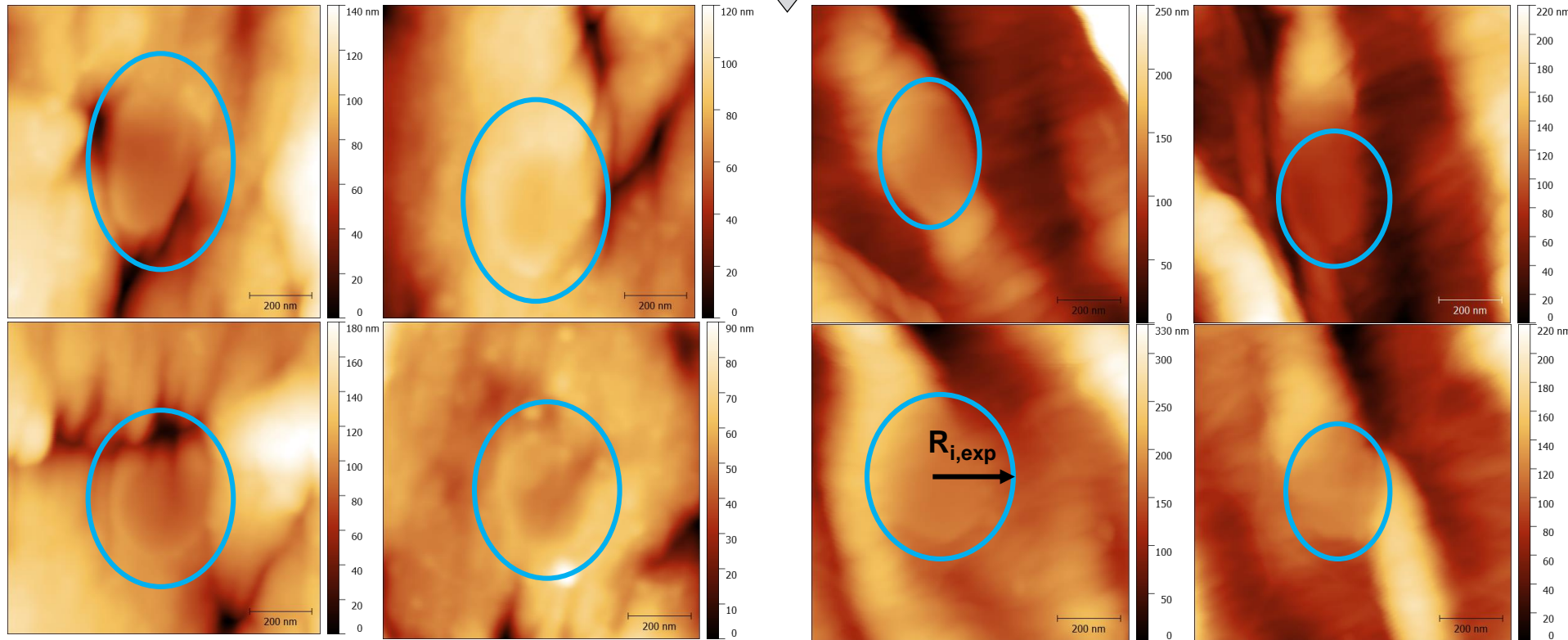
III. after unloading, the surface is different from (I) due to plastic deformation

# After AFM-NI

Cantilever orientation

Viscose fiber

Paper fiber



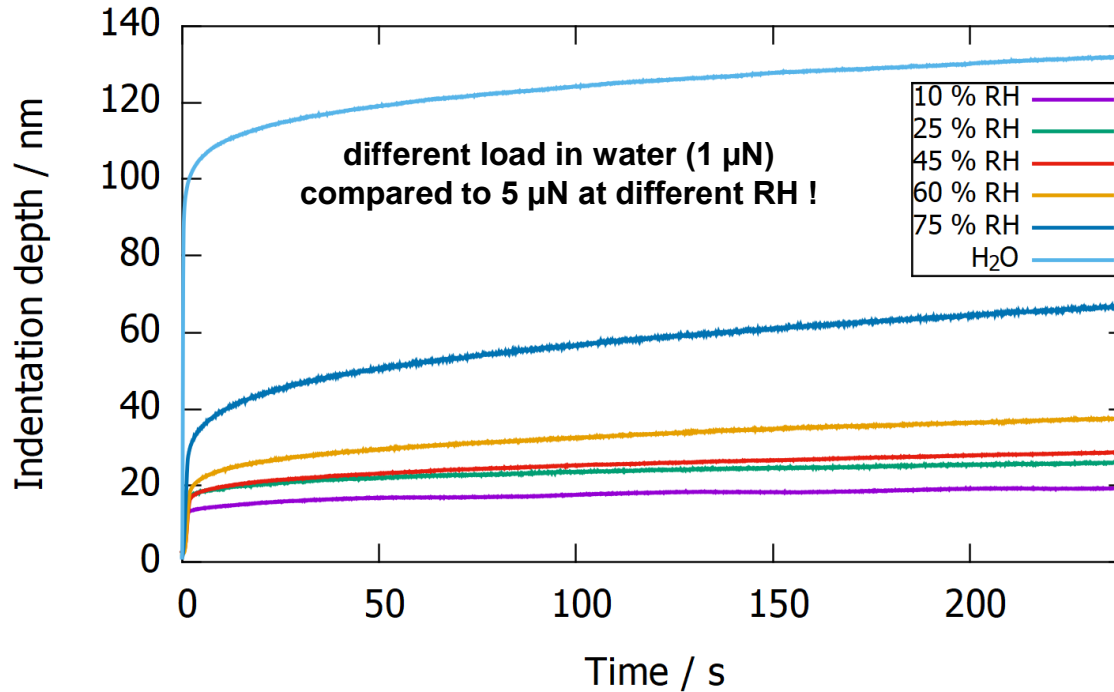
Sample	$R_{i,exp}/R_{tip}$
Viscose fiber	$8.55 \pm 5.15$
Paper fiber	$4.15 \pm 2.30$

$R_{i,exp}$  .. Experimentally measured hole radius

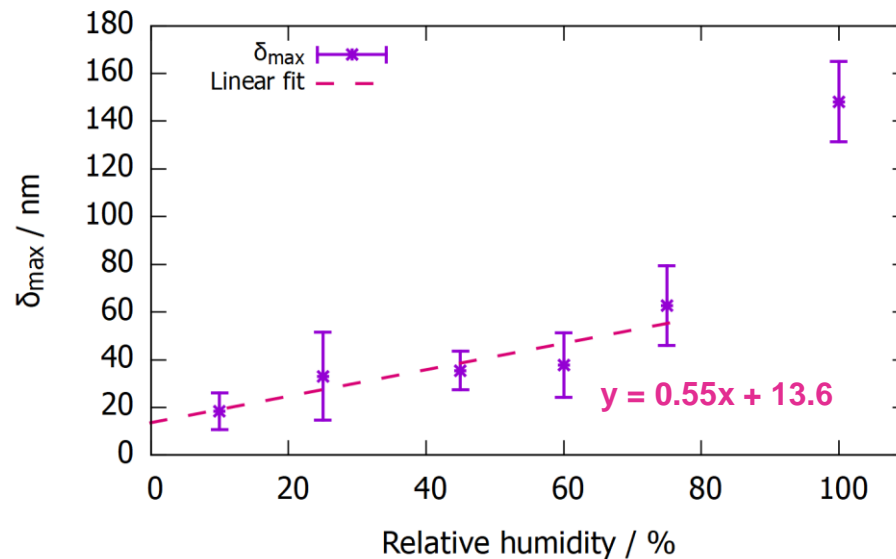
The hole radius is larger than the tip radius.

# Experimental curves

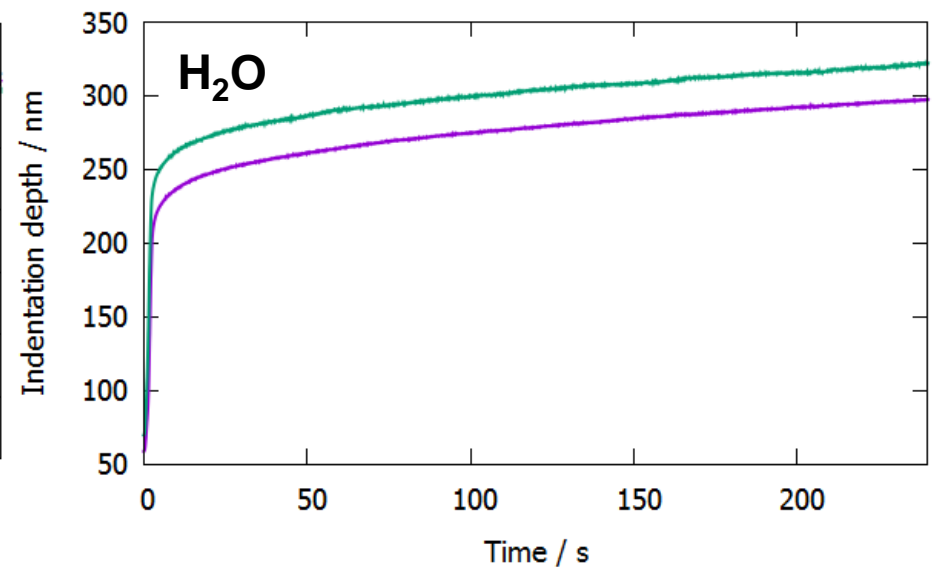
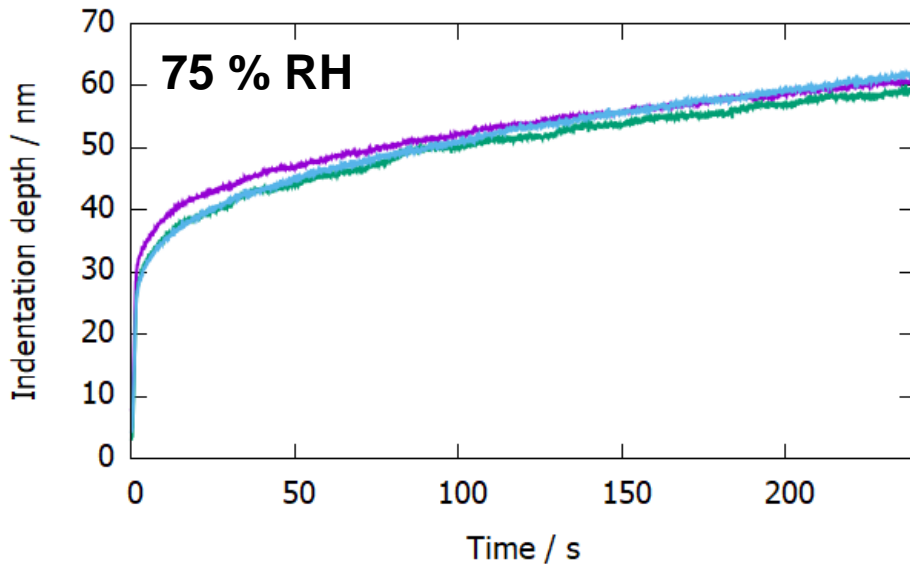
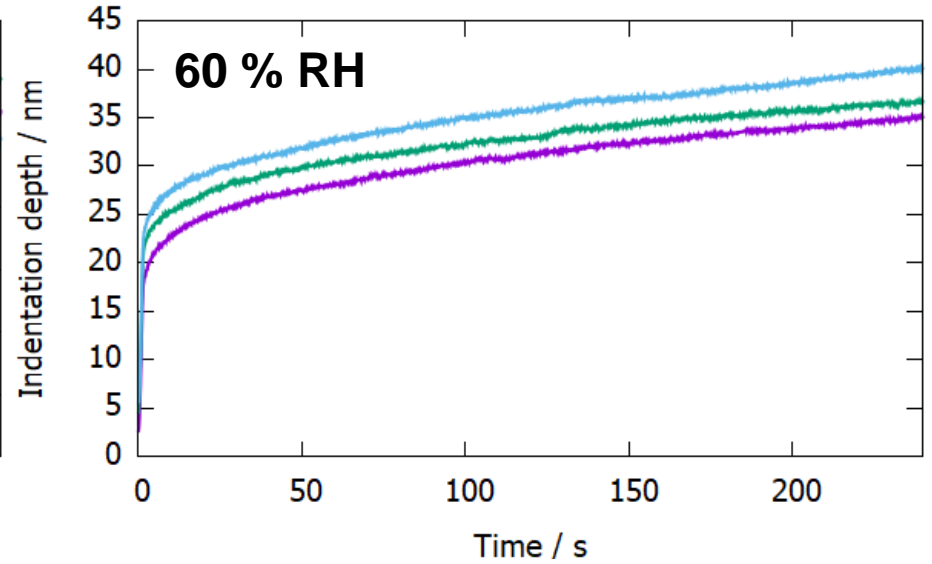
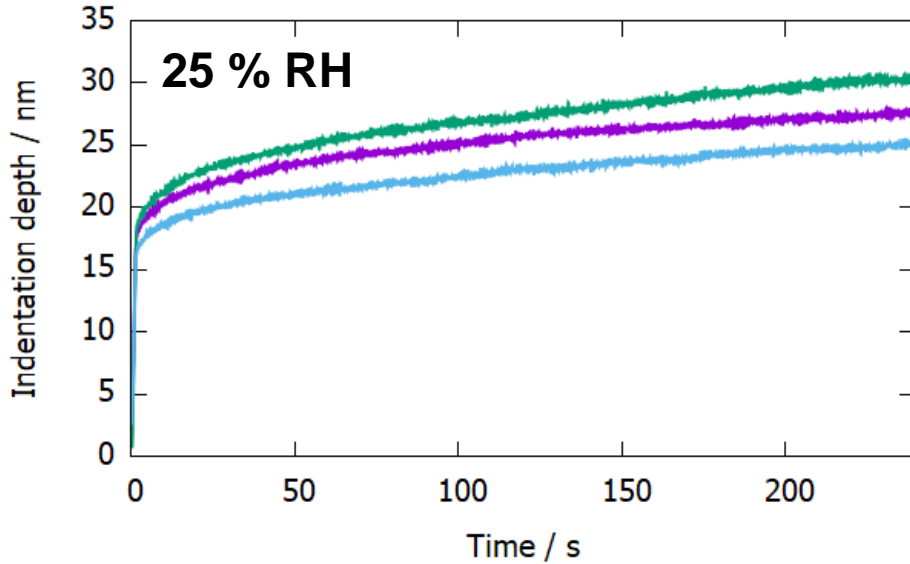
Averaged curves from 6 different Monopol X pulp fibers



**Indentation depth and slope increases with increasing RH as expected.**

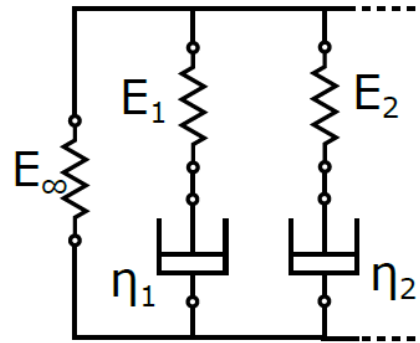


# Indentation depth at different RH



Indentation depth and slope increases with increasing RH.

# Generalized Maxwell order 2 (GM2) model



$$\sigma = A\varepsilon + B\dot{\varepsilon} + C\ddot{\varepsilon} - D\dot{\sigma} - E\ddot{\sigma}$$

**Parameters: elastic moduli & viscosities**

$$A = E_\infty \quad B = \left( \frac{\eta_1 + \eta_2}{E_\infty} + \frac{\eta_2}{E_3} + \frac{\eta_1}{E_2} \right) E_\infty$$

$$C = \eta_1 \eta_2 \frac{E_\infty + E_1 + E_2}{E_1 E_2} \quad D = \frac{\eta_2}{E_3} + \frac{\eta_1}{E_2}$$

$$E = \frac{\eta_1 \eta_2}{E_2 E_3}$$

$$\tau_i = \frac{\eta_i}{E_i}$$

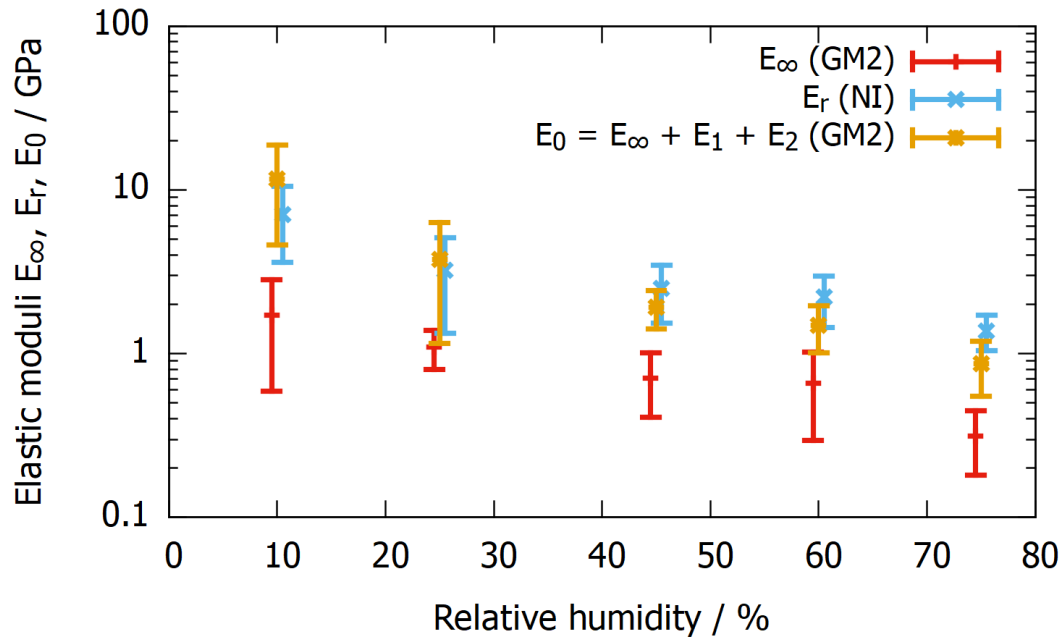
**Parameters: elastic moduli & relaxation times**

$$A = E_\infty \quad B = \left( \frac{E_1 \tau_1 + E_2 \tau_2}{E_\infty} + \tau_1 + \tau_2 \right) E_\infty$$

$$C = \tau_1 \tau_2 (E_\infty + E_1 + E_2) \quad D = \tau_1 + \tau_2$$

$$E = \tau_1 \tau_2$$

# Comparison of AFM-NI and GM2 E-moduli results



infinitely fast loading; upper limit:

$$E_0 = E_\infty + E_1 + E_2$$

infinitely slow loading; lower limit:

$$E_\infty$$

$$E_r > E_\infty$$

$$E_r \sim E_0$$

The comparison shows that  $E_0$  of the GM2 model are very close to  $E_r$  of AFM-NI whereas the  $E_\infty$  values are much lower.